Monochromator for Aberration-Corrected STEM

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In a scanning transmission electron microscope (STEM), the elemental analysis at atomic resolution is realized by the combination with electron energy-loss spectroscopy (EELS), since an aberration corrected probe forming lens system enables us to obtain an electron probe sized about 0.1 nm [1]. In EELS, the energy resolution is mainly limited by the energy spread of the electron source, which is approximately 0.7 ~ 1.0 eV for a Schottky source and 0.3 ~ 0.5 eV for a cold field emission source. For the analysis of a detailed electronic state at an atomic scale at higher energy resolution, we developed a monochromator for an aberration-corrected STEM, which integrates a double Wien-filter system. [2]

The developed monochromator, which is placed between an extraction anode of Schottky source and an accelerator tube, consists of two Wien-filters and an energy selection slit inserted between the two filters. Figure 1 shows the results of the numerical calculation of the electron trajectories from the electron source to the plane of the exit crossover of the monochromator. The electron trajectories inside of the monochromator are set to be symmetric to the slit plane so that the energy-dispersion at the slit plane is cancelled at an exit plane. Therefore, the electron probe at the specimen plane is achromatic and stigmatic. The obtained energy-spread of the electron probe is controllable independently with the probe size by choosing the width of the slit. In addition, the accelerating voltage is changeable independently on the setting of the monochromator and the electron trajectories inside the monochromator, since the accelerator of the electron gun is located after the monochromator and the potential along the optical axis inside the monochromator is kept constant. The ultimate energy resolutions of 36 meV and 30 meV were obtained with 0.1s acquisition and the slit of 0.25 μm at 200 kV and 60 kV, respectively.

Figures 2 show carbon K-edge spectra from a diamond, as an example showing the difference in the energy-loss near-edge structure (ELNES) on the different energy resolutions. Figure 2 (a) was obtained with a Schottky source and its energy resolution is approximately 1 eV. The spectra obtained with the slits of 4 μm, 2 μm and 0.5 μm at 200 kV are shown in Figs. 2 (b)-(d), respectively. With the monochromated electron source, a core exciton peak, which was unable to be detected with a standard schottky, is clearly detectable. And a dip, appeared between the main peak of K-edge and the exciton peak approximately at 289 eV, was getting deeper as the energy resolution is getting higher. In order to investigate the analytical performances with both of high energy resolution and high spatial resolution, we tried the atomic resolution EELS mapping using a SrTiO₃ specimen along the <100> direction obtained with monochromated electrons whose energy spread is 142 meV at 60kV, as shown in Fig. 3. Thus, the atomic columns of Sr and Ti in a SrTiO₃ have been successfully visualized using the monochromated electrons with 142 meV energy spread.

References:
Fig. 1. (a) Calculated trajectories along optical axis from source to exit of monochromator, (b) beam shapes at slit plane with an energy-dispersed 1st focus and (c) beam shapes at exit plane with an achromatic 2nd focus. The red lines and the green lines show the trajectories having energies of 1400 and 1401 eV inside of the monochromator.

Fig. 2. Raw spectra of carbon K-edge of diamond obtained at 200 kV, using (a) Schottky source, (b)-(d) monochromated sources with energy spreads of 294 meV, 146 meV and 50 meV, respectively.

Fig. 3. Atomic resolution EELS map of SrTiO3 [100] recorded at 60 kV with monochromated electrons of 142 meV energy spread. The probe current was 19 pA and dwell time in each pixel was 0.01 sec., (a) HAADF, (b) Sr M-map, (c) Ti L-map, (d) O K-map, (e) Overlay map.